

Rational Design of Wide Band Gap Buffer Layers for High-Efficiency Thin-Film PV

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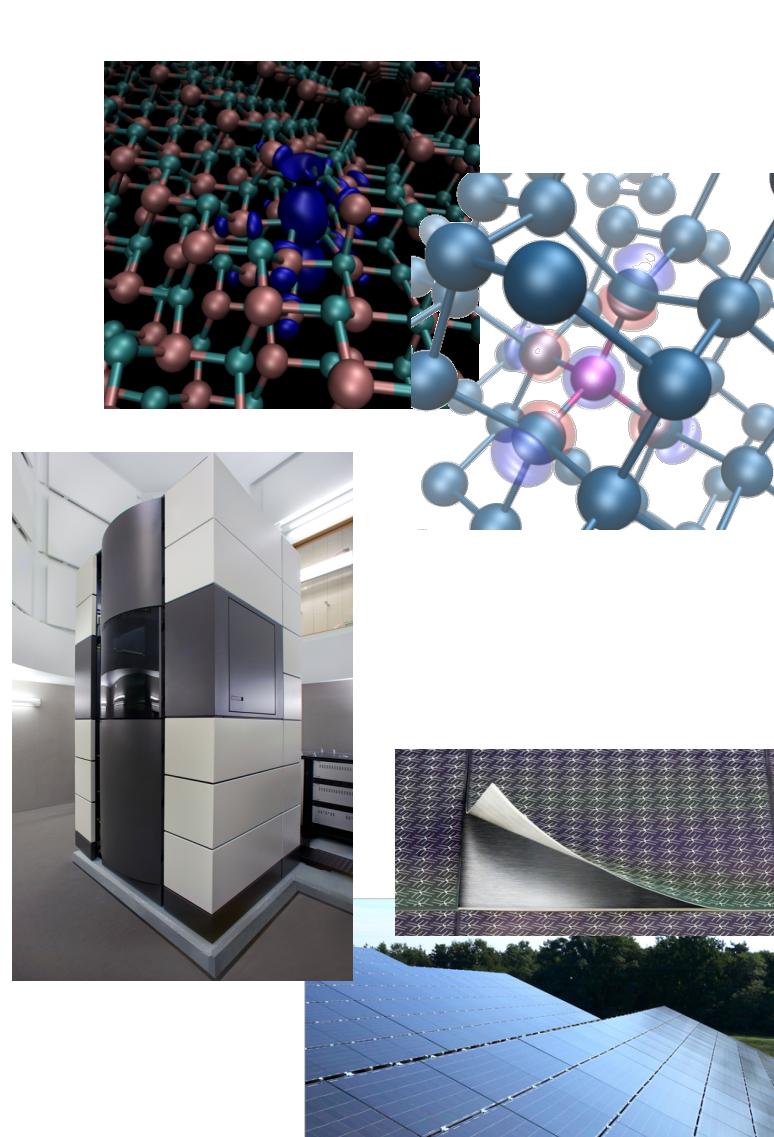
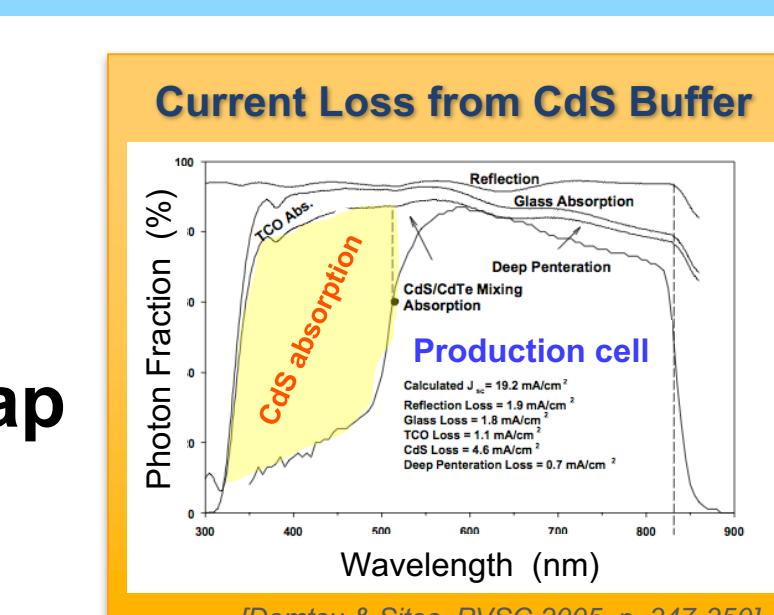
SunShot BRIDGE (Bridging Research Interactions through Collaborative Development Grants in Energy)

1. Introduction

- Efficiency of production thin-film PV is significantly impacted by absorption in buffer layer (CdS)
- Higher performance **wide band gap buffer material** is needed
- Objectives:**
 - Combine theory and characterization to enable optimization of new buffer materials
 - Focus on defects & interfaces**

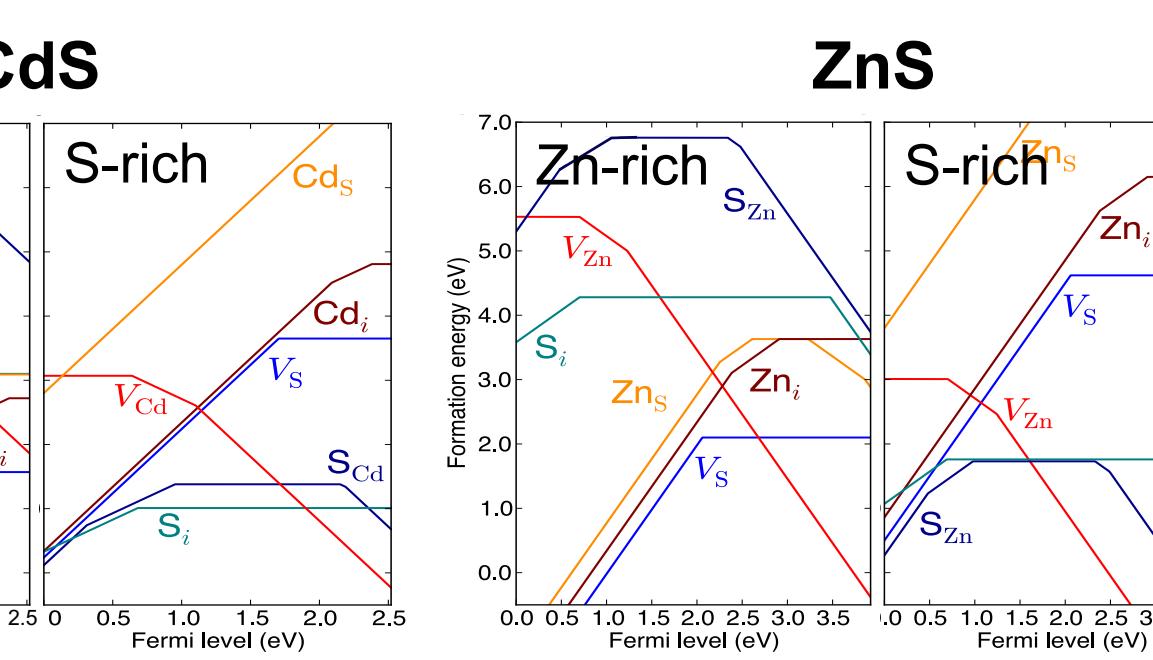
Methods

- First-principles materials simulations of defect properties using hybrid functional DFT
- Atomic-scale imaging & compositional analysis using aberration-corrected scanning transmission electron microscopy
- World-class manufactured materials from MiaSolé Hi-Tech using all-PVD roll-to-roll process



2. Theoretical Basis of Effects of Point Defects, Band Alignments, and Fermi Level Pinning

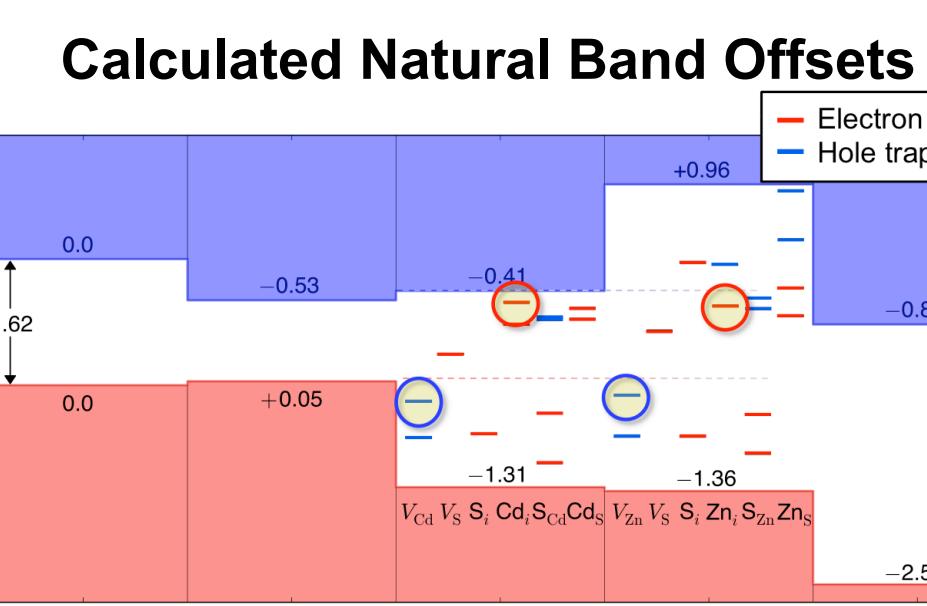
Native Defects



- Cd-rich: Abundant donors
- S-rich: V_{Zn}²⁺-compensating acceptors make n-type doping difficult
- S-rich: Abundant acceptors, n-type unachievable

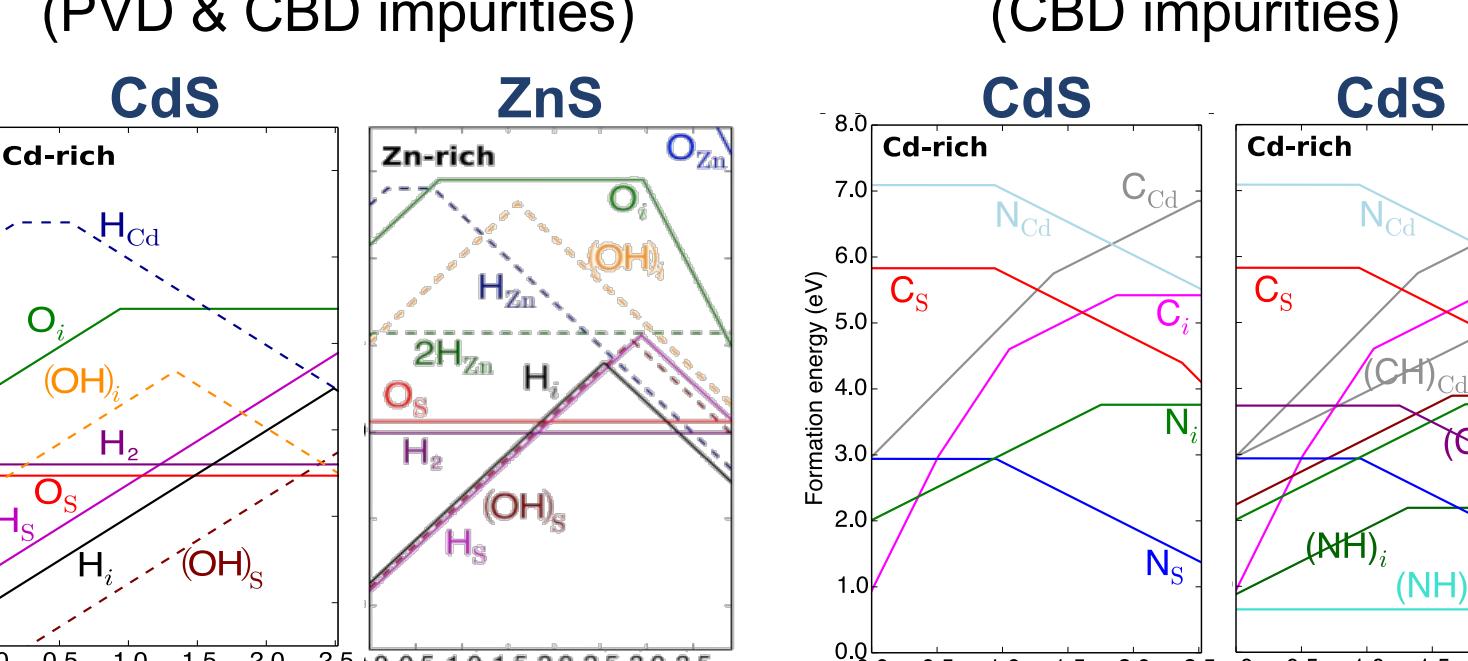
→ Large conduction band offset of ZnS makes native Zn vacancies favorable, which compensate donors

- CdS has near-ideal ΔE_{CBM} wrt Cl(GS)
 - ZnS CBM too high!
 - ZnO CBM too low, but large E_g opened in VBM
- ⇒ (Cd,Zn)(O,S) can be ideal



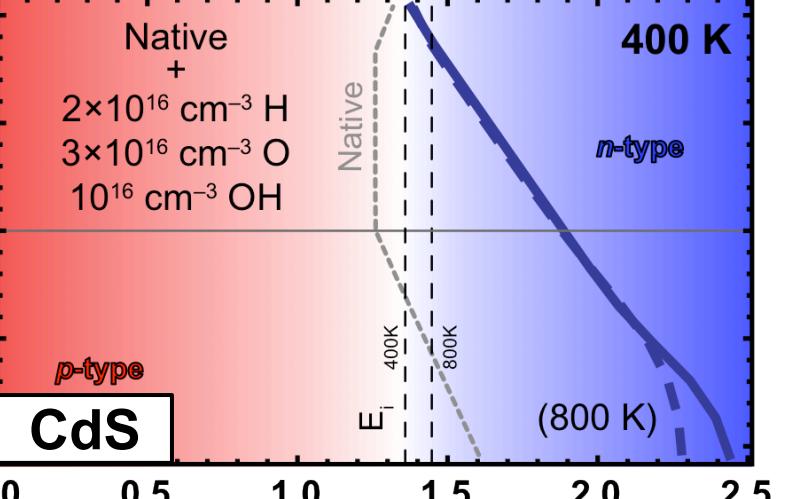
Growth-Related Impurities

O, H, & OH defects (PVD & CBD impurities)

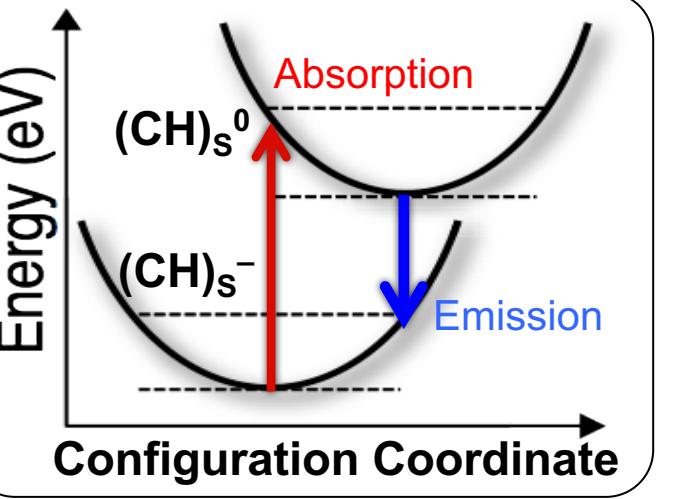


- O, H, OH in CdS are shallow donors → aid n-type doping
- In ZnS, these are amphoteric and pin Fermi level more intrinsic

Calc'd equilibrium Fermi level



Optical absorption from C-H defects is in the visible



Intermixing

Migration Energy Barriers (eV)

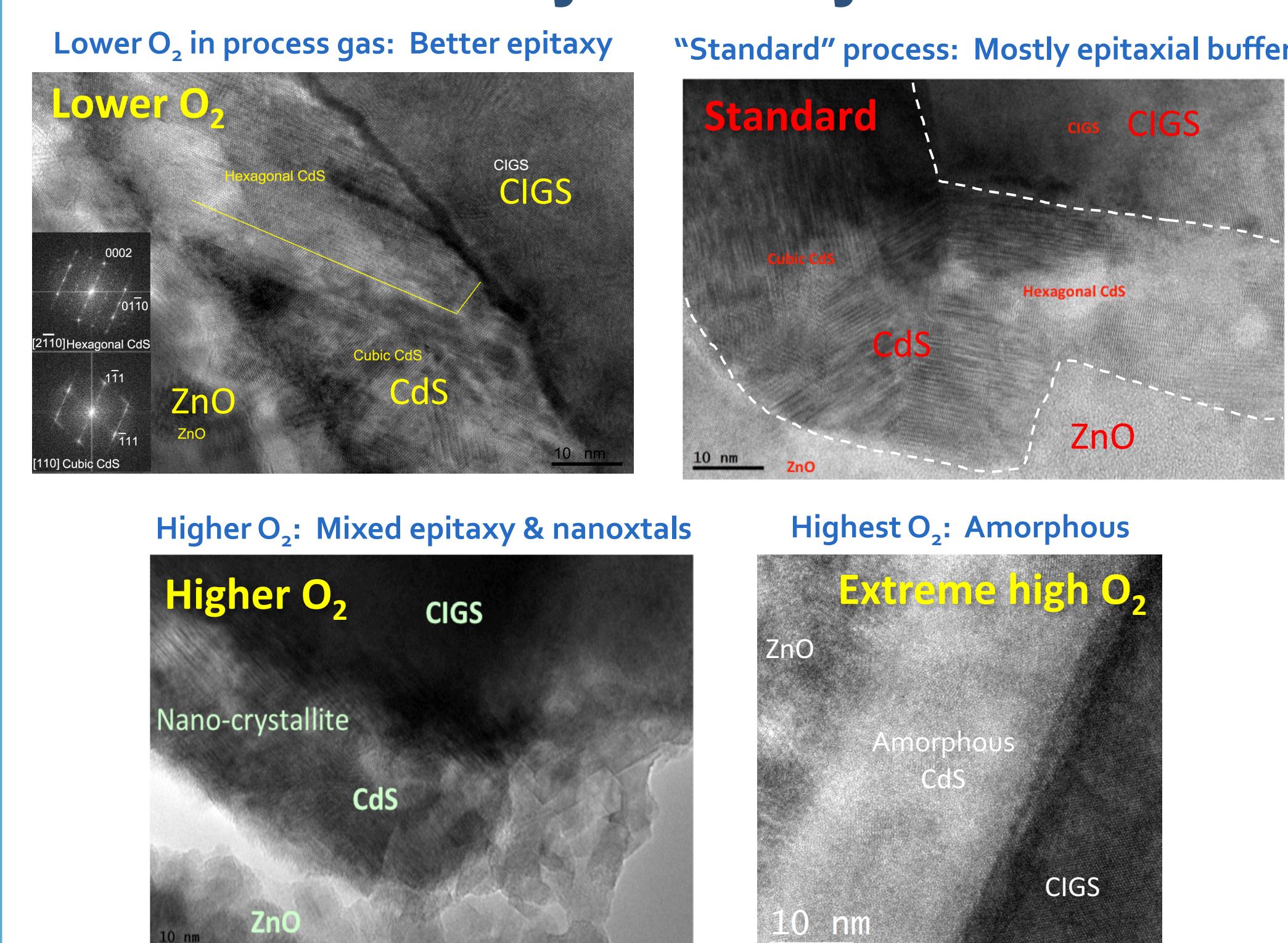
Defect	CdS c-axis	CdS ⊥ c-axis	ZnS c-axis	ZnS ⊥ c-axis
V _{Zn} ²⁺	1.02	1.02	1.21	1.23
V _S ⁰	2.44	2.07	2.90	2.49
In ⁺	0.73	0.51	0.66	0.38
Ga ⁺	0.94	0.75	0.95	0.87
Se ⁰	0.40	0.38	0.47	0.33
Cu ⁺	0.64	0.26	0.30	0.23
Zn ⁰	0.69	0.33	0.59	0.44
Zn ²⁺	0.64	0.27	0.44	0.32
Sn ²⁺	1.13	1.00	1.19	0.92
Na ⁺	0.99	1.09	1.51	1.82
K ⁺	1.51	0.50	1.49	0.45

[J.B.Varley & V.L., *J. Appl. Phys.*, in prep (2014).]

Interface also may comprise a number of interphase compounds:

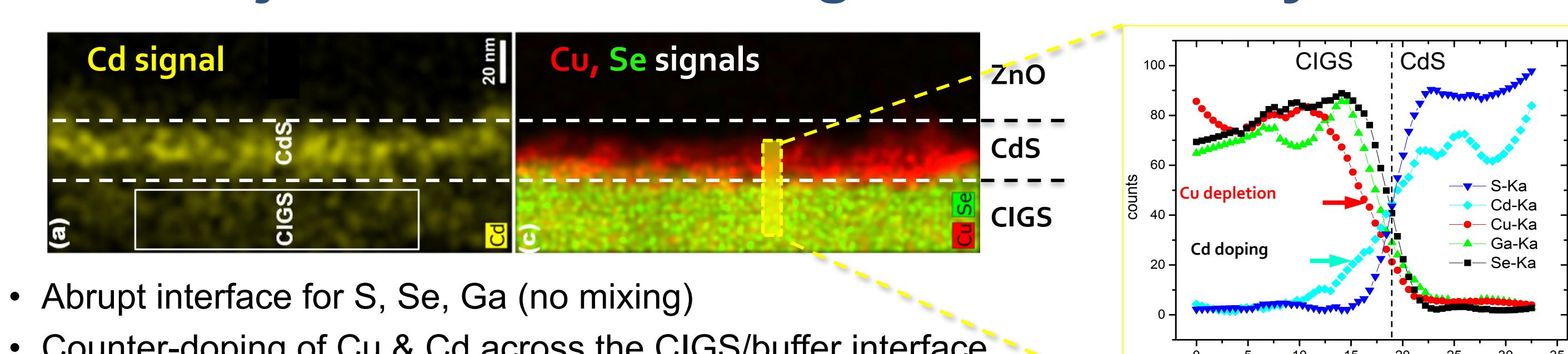
3. Control of Film Deposition Characterized by STEM Imaging and EDS Analysis

Crystallinity

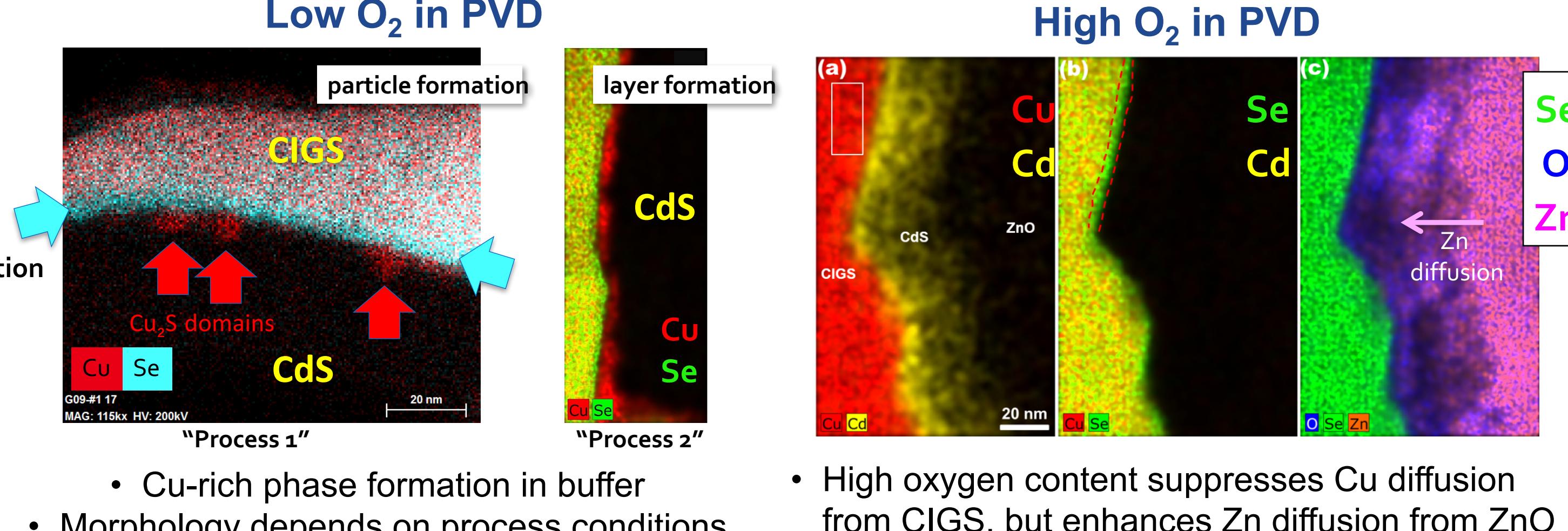


- O₂ in process gas modifies lattice match and crystallinity
- Can achieve epitaxial CdS with PVD sputter process

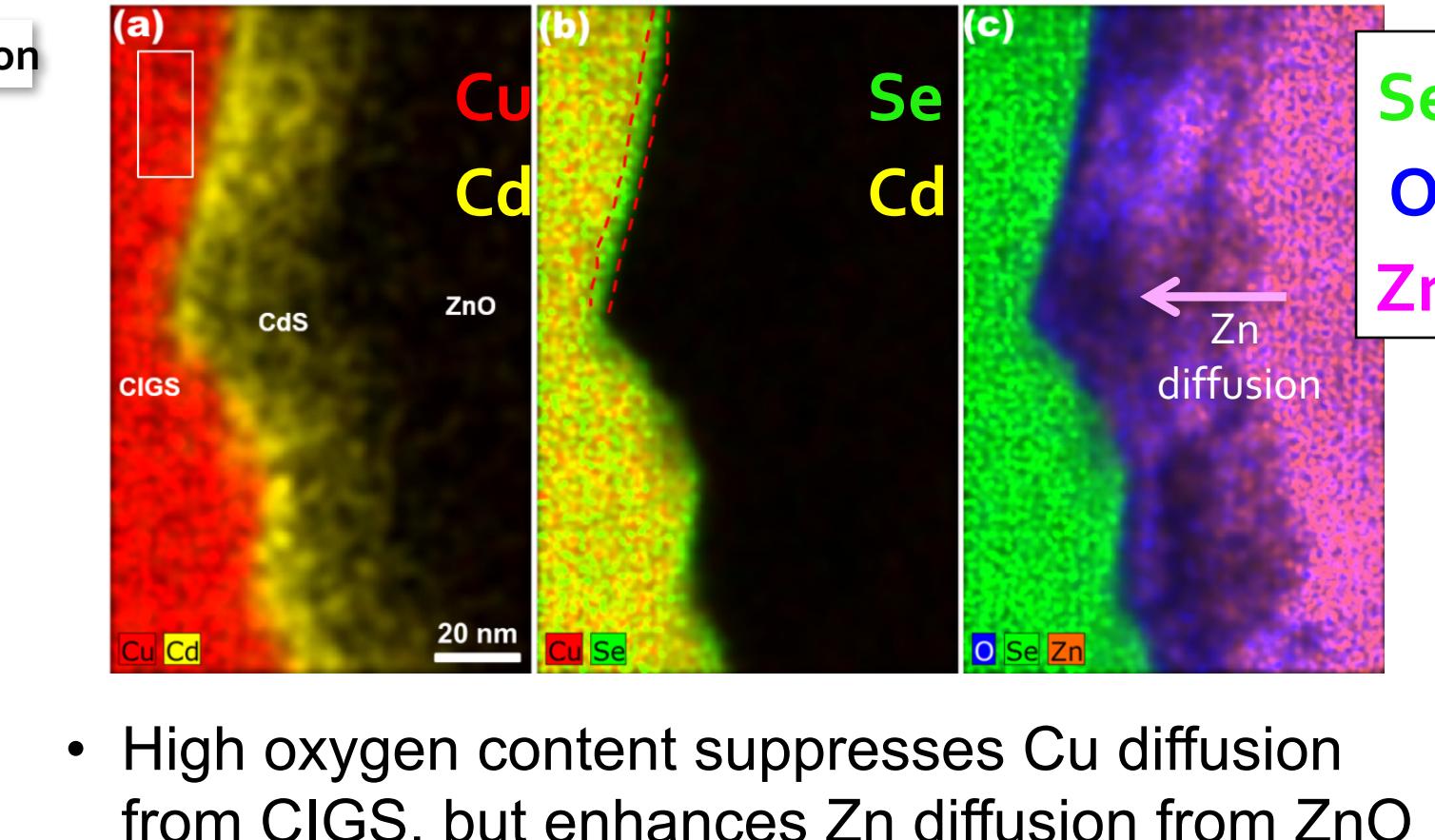
Heterojunction Intermixing and Secondary Phases



Low O₂ in PVD



High O₂ in PVD

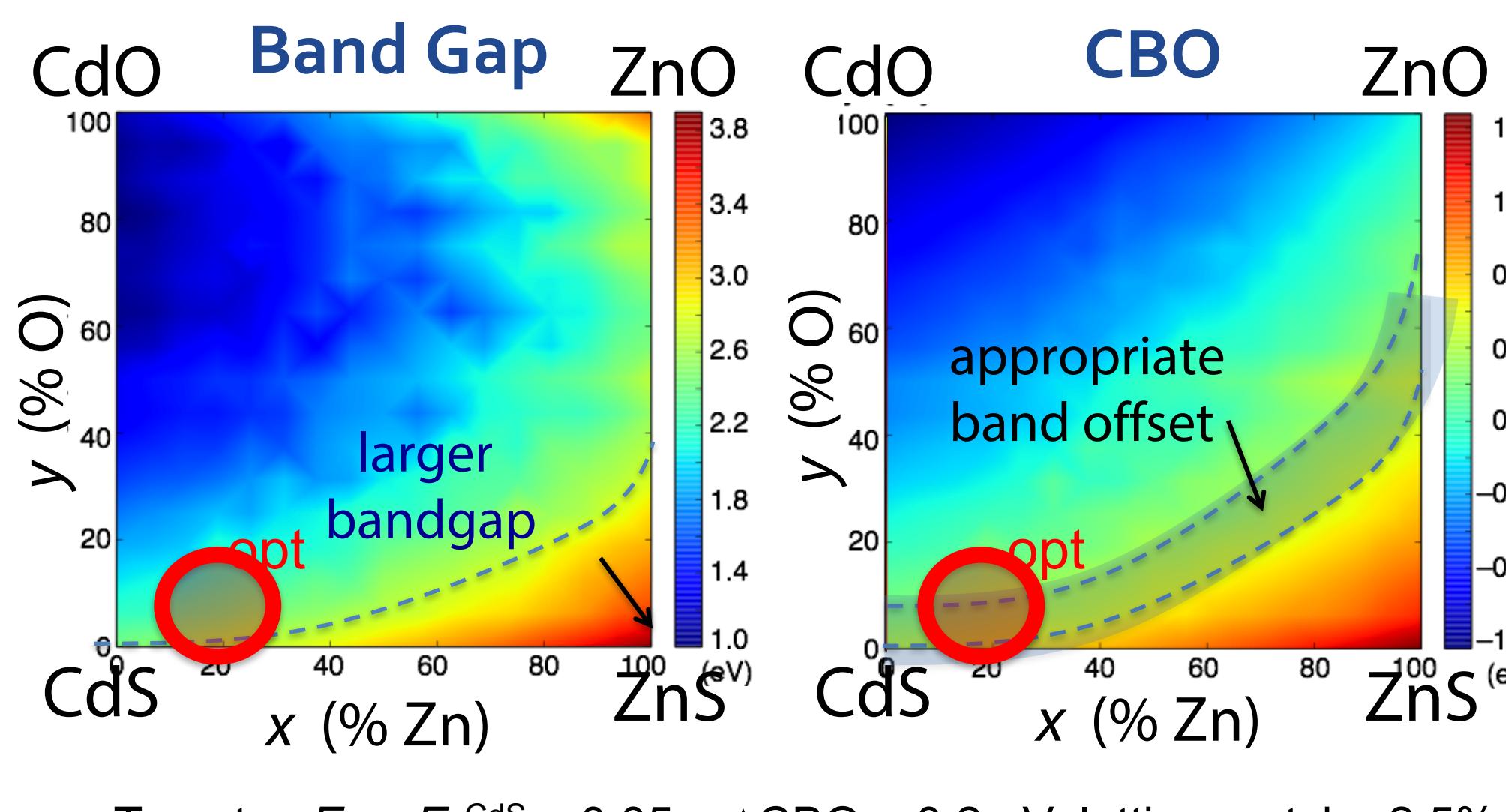


4. Computational Optimization of Alloy Composition for High-Performance Buffer

(Cd_{1-x}Zn_x)(O_yS_{1-y}) quaternary provides optimization degrees of freedom

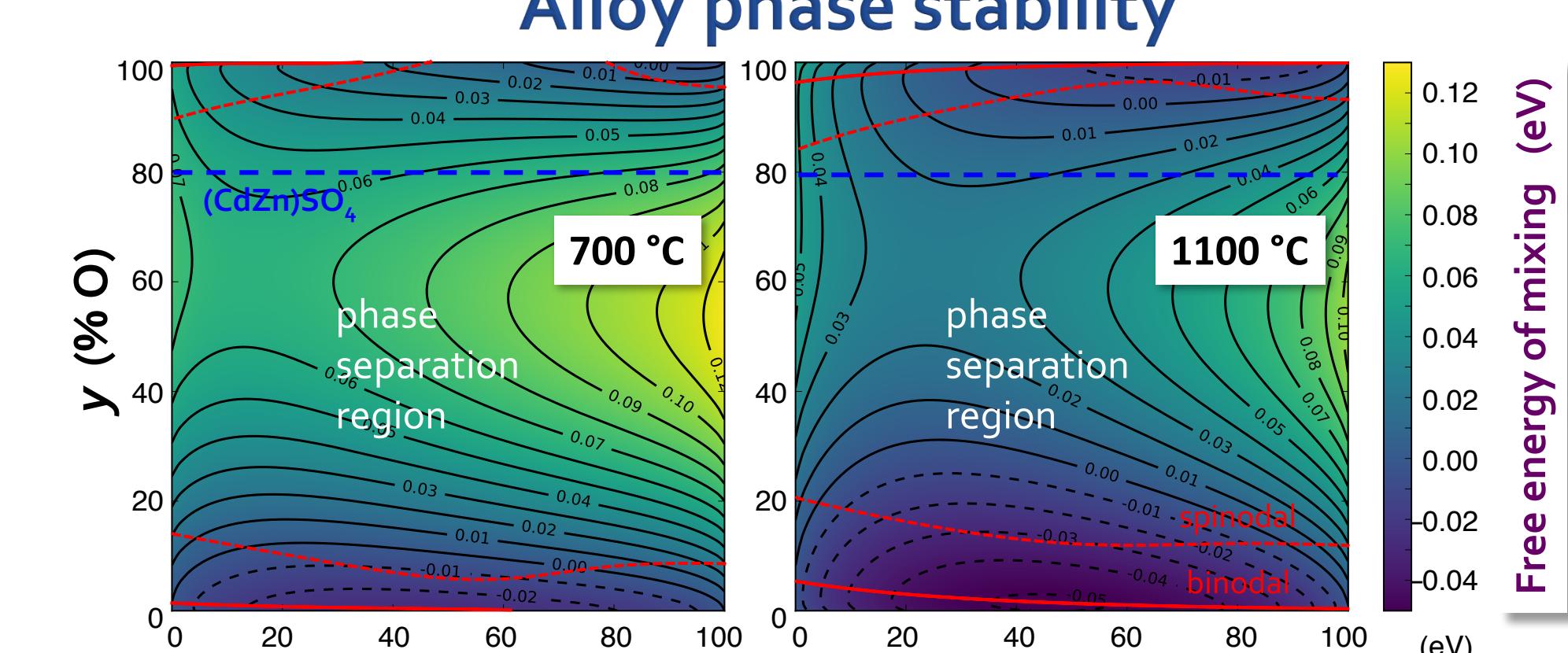
Properties:

- band gap
- band offset
- lattice match
- dopability



- Targets: $E_g > E_g^{\text{CdS}}$, $-0.05 < \Delta \text{CBO} < 0.2 \text{ eV}$; lattice match $\pm 2.5\%$

Alloy phase stability



- Thermodynamic driving force for phase separation even at high T
- Secondary phases are very stable: e.g. (Cd,Zn)SO₄

5. Major Conclusions

- CdS is naturally favorable for its n-type dopability and favorable band alignment with CIGS
- ZnS is a BAD wide band gap buffer: ΔE_c too high, hard to n-dope, recombination centers from native defects
⇒ **(Cd,Zn)(O,S) is a promising alloy system**
- CIGS/buffer interface properties are critical and complicated
 - Recombination affected by crystallinity
 - Intermixing can bury p-n junction (Cd counter-doping)
 - Intermixing can lead to unfavorable secondary phases (Cu in CdS)
 - Sputter deposition conditions, particularly oxygen in process gas, can control quality and composition of interface region
- Careful selection and optimization of buffer material is dependent on absorber material

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